

CRYSTAL STRUCTURE AND ELECTRICAL CONDUCTIVITY OF A NEW
CONDUCTING BISDITHIOLATE COMPLEX $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$

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The X-ray crystal structure analysis of $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ (mnt = 1,2-dicyanovinylene-1,2-dithiolato, $\text{C}_4\text{N}_2\text{S}_2^{2-}$) has revealed that the $[\text{Pt}(\text{mnt})_2]^{0.5-}$ anions are stacked face-to-face along the a-axis to form a fourfold platinum chain. In contrast to the regular chain in the metallic compound $\text{Li}_{0.8}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$, the fourfold structure can be regarded as a Peierls insulating structure. The room temperature conductivity is about 1 S cm^{-1} .

Recently a new type of one-dimensional(1-D) metallic complex containing $[\text{M}(\text{mnt})_2]^{n-}$ (M=Ni, Pd, Pt) anions has aroused considerable interest. The first observation of metallic behavior due to interacting $[\text{Pt}(\text{mnt})_2]^{n-}$ anions was reported for single crystals of $\text{Li}_{0.8}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ ^{1,2)} whose room temperature conductivity is 200 S cm^{-1} . The $[\text{Pt}(\text{mnt})_2]^{n-}$ anions stack face-to-face with a Pt··Pt distance of 3.639 Å. This is longer than that found in the insulating Pt^{II} complex $\text{K}_2[\text{Pt}(\text{CN})_4] \cdot 3\text{H}_2\text{O}$ ³⁾ (3.480 Å) which has the room-temperature conductivity of $10^{-8} \text{ S cm}^{-1}$. It is well-known that the Pt··Pt distance is shortened by the partial oxidation of the complex to form the metallic band along the Pt chain ($r_{\text{Pt} \cdots \text{Pt}} = 2.89 \text{ Å}$ ($\text{K}_2[\text{Pt}(\text{CN})_4]\text{Br}_{0.3} \cdot 3.2\text{H}_2\text{O}$ ⁴⁾), 2.80 Å ($\text{Rb}_2[\text{Pt}(\text{CN})_4](\text{FHF})_{0.4}$ ⁵⁾), 2.96 Å ($\text{K}_{1.75}[\text{Pt}(\text{CN})_4] \cdot 1.5\text{H}_2\text{O}$ ⁶⁾)). The large Pt··Pt distance of $\text{Li}_{0.8}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ indicates that the conduction pathway is not formed by the overlap of the $5d_{z^2}$ orbitals of the Pt atoms, but by intermolecular overlap of delocalized molecular orbitals of the whole anions in which the atomic orbitals of S in the ligands play a prominent role. Thus $\text{Li}_x[\text{Pt}(\text{mnt})_2] \cdot n\text{H}_2\text{O}$ occupies a position in-between the 1-D platinum complexes and the sulfur donor organic metals.

Recently we have prepared conducting bisdithiolate complex $\text{Li}_x[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$. X-ray examination of the crystals revealed two types of crystals. The needle-shaped crystals have previously been identified as $\text{Li}_{0.8}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$. We report here the crystal structure and the anisotropic electrical conductivity of the newly found elongated plate crystals $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$.

tances vary from 2.127 Å to 2.629 Å. The fourfold structure of $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ corresponds to the $2k_F$ (k_F : Fermi wave number) Peierls structure.⁹⁾ The resistivities of $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ were measured by 4-probe technique (Fig. 2). The room temperature conductivities along the stacking a-axis and the direction of $[\bar{1}10]$ are almost the same ($\approx 1 \text{ S cm}^{-1}$). The small anisotropy or 2-D character of the conductivities in (001) plane is in a striking contrast to the strong 1-D character of $\text{K}_2[\text{Pt}(\text{CN})_4]\text{Br}_{0.3} \cdot 3.2\text{H}_2\text{O}$, whose conductivity along the chain direction is 10^5 times larger than the transverse conduction. The temperature dependence of the conductivity shows the compound to behave as a semiconductor. The conductivity is lowest along the c-axis ($\approx 10^{-3} \text{ S cm}^{-1}$).

It is well-known that the electrical conductivity of 1-D platinum complexes such as tetracyanoplatinates and bis(oxalato)platinates is strongly dependent on the periodicity of the Pt chain. Similar correlation between electrical properties and the periodicity of the columns can be seen in $[\text{Pt}(\text{mnt})_2]$ complexes. As mentioned before, $\text{Li}_{0.8}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ is metallic at room temperature but it undergoes metal-insulator transition at 213 K (T_c). Since the complexes are stacked regularly above T_c , the band is partially (41%) filled and 1-D metallic properties are observed. The $2k_F$ lattice distortion waves begin to develop below T_c , which open the energy gap at $k=k_F$ (k_F : Fermi wave number) and the compound transforms to a semiconducting state. In $\text{Rb}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$, the band is half filled and a Peierls instability tends to produce a dimerization of the unit cell which is consistent with the dimer structure of $\text{Rb}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$. Since the period of the $2k_F$ (Peierls distortion) wave is strongly commensurate with the spacing of the $\text{Pt}(\text{mnt})_2$ complexes the band gap will be large and the electrical conductivity

may be expected to be small. Experimentally the room temperature conductivity of $\text{Rb}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ is found to be very small ($\approx 10^{-5} \text{ S cm}^{-1}$). In the case of $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ the band is 1/4 filled and the Peierls distortion produce a fourfold structure with a band gap at the Fermi level resulting in semi-conducting properties which agree with the temperature dependence of the resistivity described above.

Although $\text{M}_x[\text{Pt}(\text{mnt})_2]$ has the periodicity of the stacks characteristic of 1-D conducting platinum complexes, this complex cannot be regarded as a typical 1-D conductor. In an ideal 1-D system fluctuations play an essential part and the relationship between band gap (2Δ) and the transition temperature (T_c) for a system without fluctuations $2\Delta/T_c=3.73$, cannot be applied. In addition in a 1-D system the $2k_F$ lattice distortion wave does not develop until the temperature is lowered to as low as $T \lesssim T_c/3$ ($2\Delta/T_c \gtrsim 10$ (1-D system)). However, in $\text{Li}_{0.8}$ -

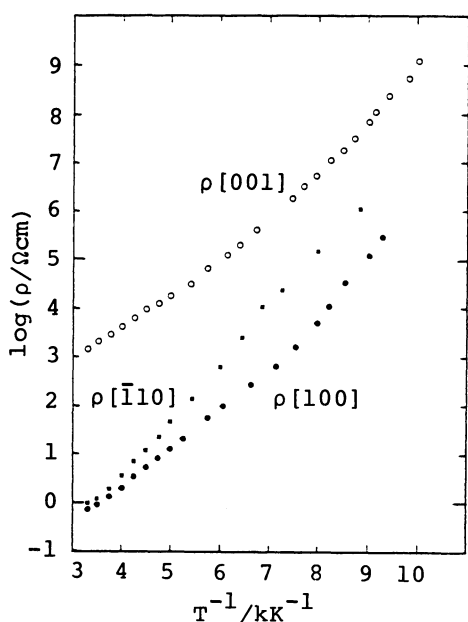


Fig. 2. The resistivity of $\text{Li}_{0.5}[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$.

$[\text{Pt}(\text{mnt})_2] \cdot 2\text{H}_2\text{O}$ the activation energy of the electrical conduction (Δ) and the transition temperature (T_c) gave the value $2\Delta/T_c$ of 3.95, which is close to the value derived from the mean field theory 3.73 and the lattice distortion developed close to T_c . The reason for the depression of fluctuations of the lattice distortion wave exists in the characteristic inter-chain transverse contacts *via* the S atoms, which are commonly observed in $M_x[\text{Pt}(\text{mnt})_2]$ complexes. Similar transverse contacts can be also seen in the organic conductors where the S·S contacts play an essential role for the formation of the conduction pathway (BEDT-TTF₂X, X=ClO₄, PF₆..).¹⁰⁾ The transverse arrays of molecules observed in these conductors will afford the way to design new 2-D conductors, which will be important for the synthesis of conducting materials which do not exhibit the Peierls instability.

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References

- 1) A.E.Underhill and M.M.Ahmad, J. Chem. Soc., Chem. Commun., 1981, 67.
- 2) A.Kobayashi, Y.Sasaki, H.Kobayashi, A.E.Underhill, and M.M.Ahmad, J. Chem. Soc., Chem. Commun., 1982, 390.
- 3) D.M.Washecheck, S.W.Peterson, A.H.Reis, Jr., and J.M.Williams, Inorg. Chem. 15, 74 (1976).
- 4) G.Heger, H.J.Deiseroth, and H.Schulz, Acta Crystallogr., Sect.B,34, 725 (1978).
- 5) A.J.Schultz, C.C.Coffey, G.C.Lee, and J.M.Williams, Inorg. Chem., 16, 2129 (1977).
- 6) J.M.Williams, K.D.Keefer, D.M.Washecheck, and N.P.Enright, Inorg. Chem., 15, 2446 (1976); A.H.Reis, Jr., S.W.Peterson, D.M.Washecheck, and J.S.Miller, Inorg. Chem., 15, 2455 (1976).
- 7) M.M.Ahmad, D.J.Turner, A.E.Underhill, A.Kobayashi, Y.Sasaki, and H.Kobayashi, J. Chem. Soc., Dalton Trans., in press.
- 8) A.Kobayashi and Y.Sasaki, Bull. Chem. Soc. Jpn., 50, 2650 (1977).
- 9) H.Kobayashi, I.Shirotnani, A.Kobayashi, and Y.Sasaki, Solid State Commun., 23, 409 (1977).
- 10) H.Kobayashi, A.Kobayashi, Y.Sasaki, G.Saito, T.Enoki, and H.Inokuchi, J. Am. Chem. Soc., 105, 297 (1983); H.Kobayashi, T.Mori, R.Kato, A.Kobayashi, Y.Sasaki, G.Saito, and H.Inokuchi, Chem. Lett., 1983, 581.

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